



Growth of 2D MoP single crystals on liquid metals by chemical vapor deposition

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ABSTRACT Two-dimensional (2D) transition metal phosphides (TMPs) are predicted with many novel properties and various applications. As a member of TMPs family, molybdenum phosphide (MoP) exhibits many exotic physico-chemical properties. However, the synthesis of high-quality 2D MoP single crystals is not reported due to the lack of reliable fabrication method, which limits the exploration of 2D MoP. Here, we report the growth of high-quality ultrathin MoP single crystals with thickness down to 10 nm on liquid metals *via* chemical vapor deposition (CVD). The smooth surface of liquid Ga is regarded as a suitable growth substrate for producing 2D MoP single crystals. The Mo source diffuses toward the Ga surface due to the high surface energy to react with phosphorus source, thus to fabricate ultrathin MoP single crystals. Then, we study the second harmonic generation (SHG) of 2D MoP for the first time due to its intrinsic non-centrosymmetric structure. Our study provides an new approach to synthesize and explore other 2D TMPs for future applications.

Keywords: 2D materials, transition metal phosphides, molybdenum phosphides, liquid metals, chemical vapor deposition

INTRODUCTION

Researches in two-dimensional (2D) materials have increasingly grown triggered by pioneering studies on graphene [1–4]. As an emerging family of 2D materials, 2D transition metal phosphides (TMPs) are predicted with novel properties, which are appealing for many applications, including optical, thermal insulating materials, magnetic material and catalysts [5,6]. Thus the reliable synthesis of 2D TMPs is highly demanded.

As a member of TMPs family, molybdenum phosphide (MoP) exhibits many exotic properties with the presence of novel three-component fermions, such as intrinsic non-centrosymmetric structure, high conductivity, strong anisotropic lattice thermal conductivity, superconductivity, and the transformation to a direct bandgap semiconductor in monolayer MoP [7–11]. Yet, traditional solution approaches usually obtain polycrystalline nanoparticles [12,13]. The preparation of high-quality 2D MoP single crystals has not yet been reported due to the lack of reliable fabrication methods. Chemical vapor deposition (CVD) approach is considered to be one of the most promising methods to obtain high-quality 2D single crystals [14]. The emerging liquid metals possess atomically smooth surface, which are regarded as suitable growth substrates for producing 2D materials [15].

In this study, we firstly report the successful growth of high-quality 2D MoP single crystals *via* CVD by using liquid gallium (Ga) surface as the growth substrate. Our group has successfully prepared a series of 2D materials on liquid metal surface [16–19]. Mo source supplied by Mo foil can diffuse toward the Ga surface due to the high surface energy. This enables a well-designed chemical reaction with phosphorus source on the Ga surface, thus to fabricate ultrathin MoP single crystals. Due to the intrinsic non-centrosymmetric structure, MoP would exhibit second harmonic generation (SHG), which is an essential building block for future laser optics, optical communications, and photonic circuits [20]. Therefore, we explore the SHG property of the as-prepared high-quality 2D MoP single crystals for the first time, which reveals their potential as ultrathin frequency doubling

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crystals in nonlinear optical (NLO) fields. We hope this study of growing high-quality 2D MoP single crystals on liquid metals would provide an approach to synthesize and explore many other 2D TMPs for future applications.

EXPERIMENTAL SECTION

CVD growth of 2D MoP single crystals on liquid Ga

A commercial gallium (Ga) pellet (Alfa Aesar, >99.9% purity) was departed into small Ga droplets in hot ethanol. Molybdenum (Mo) foil (Alfa Aesar, 99.95 wt.% purity) substrate was rinsed sequentially in acetone, ethanol and ultrapure water. A Ga droplet was put on the Mo foil (1 cm×1 cm) to get Ga–Mo substrate, which was then heated in a quartz tube to 1100°C with a rate of 30°C min⁻¹. After that, red phosphorus powder (Alfa Aesar, >99.9% purity) was introduced to the tube furnace and the temperature was maintained at 500°C. The growth process was remained for 30 min under the flow of Ar (200 sccm) and H₂ (100 sccm). At last, the substrates with grown MoP samples were rapidly cooled to room temperature.

Transferring MoP to the target substrates

The substrate with the as-grown MoP was firstly spin-coated with a polymethyl methacrylate (PMMA) film. Then, PMMA/MoP film was released *via* soaking Ga in the hydrochloric (HCl) solution (the volume ratio of HCl: H₂O is 1:4) for 4 h. After that, the film was repeatedly washed in the ultrapure water to get rid of the etchant and residues. Afterward, PMMA/MoP film can be transferred to various target substrates, such as 300 nm SiO₂/Si substrates and copper (Cu) TEM grid. Finally, PMMA layer was removed by hot acetone.

Raman and SHG measurements

Raman spectrum was acquired using a WITec Alpha300 R Raman system at an excitation wavelength of 532 nm. The Raman peak of Si at 520 cm⁻¹ was served as the reference for wavenumber calibration in this characterization. SHG measurements were based on the WITec Alpha300 R confocal Raman system using an excitation source of chameleon Ti: Sapphire femtosecond tunable laser (≈140 fs, 80 MHz). The pulse laser was focused on the sample *via* objective lens and the SHG signals were then collected in reflection mode by the same objective, which was then passed through a beam splitter and short-pass edge filter ($\lambda = 750$ nm). The fundamental laser was filtered out and the SHG signal was eventually detected by the spectrometer and charge coupled device (CCD)

camera. The pulse laser was linearly polarized along horizontal direction. In order to avoid the influence of dichroic beam splitter on laser polarization ratio and polarized SHG signals, the sample was rotated with a step of 10° for the angle-resolved SHG measurement. With the help of a tunable analyzer on the collection beam path, the polarized SHG signal was then collected with either parallel or perpendicular configurations and then plotted out with the function of excitation angles.

Density functional theory (DFT) calculations

The second-order NLO susceptibility of MoP was calculated to reveal the optical property of MoP. First-principles calculation with sum-over-state (SOS) approximation was performed to calculate the frequency-dependent SHG susceptibility. For the electronic structure, the first-principles calculation based on DFT was conducted within a primitive cell with a 40 × 40 × 40 *k*-point grid and 500 eV energy cutoff. Projector augmented wave pseudopotentials with Perdew, Burke, and Ernzerhof (PBE) exchange correlation were utilized in this process.

Characterizations

The obtained samples were examined using scanning electron microscopy (SEM) (ZEISS Merlin Compact SEM), X-ray photoelectron spectroscopy (XPS) (Thermo Scientific, ESCALAB 250Xi, monochromatic Al K α radiation), atomic force microscopy (AFM) (NT-MDT Ntegra Spectra). Transmission electron microscopy (TEM) images were recorded on an aberration-corrected high-resolution TEM system (FEI Titan Themis 200, operating voltage of 80 kV) and a probe-corrected high-resolution TEM (HRTEM) system (Titan Probe corrected TEM, Titan G2 60-300, operating voltage of 300 kV). The scanning TEM (STEM) were recorded by probe Cs-corrected TEM (FEI Titan ChemiSTEM, operating voltage of 200 kV) with the high-angle annular dark-field (HAADF) and the bright-field (BF) images. Meanwhile, the elemental mapping of samples was collected by energy dispersive X-ray spectroscopic (EDX).

RESULTS AND DISCUSSION

High-quality MoP single crystals were synthesized by a CVD method. A piece of Mo foil was used to support the Ga substrate and provide Mo source, while red phosphorus served as phosphorus source (Fig. 1a). The high growth temperature allows the Mo source to diffuse toward the liquid metal Ga surface due to the high surface energy [15]. Then, Mo reacts with phosphorus on the Ga

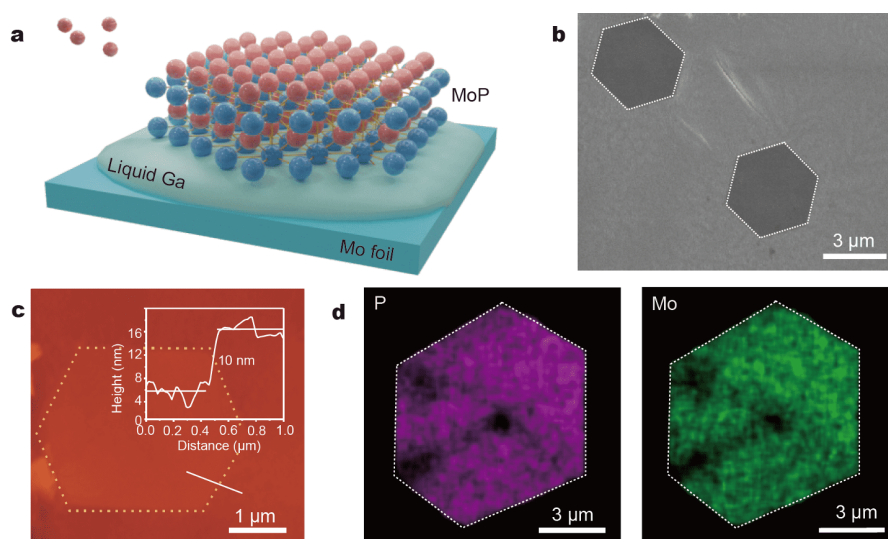


Figure 1 Morphology and chemical composition of 2D MoP single crystals. (a) Schematic of the CVD approach for synthesizing 2D MoP single crystals at ambient pressure. (b) A typical SEM image of the 2D MoP crystals. (c) A typical AFM image of the 2D MoP crystal. The inset shows the corresponding thickness profile corresponding to the region marked by a white line. (d) EDX elemental mapping of P and Mo of a hexagonal MoP single crystal.

surface, thus to fabricate ultrathin MoP single crystals (Fig. S1). SEM and AFM characterizations were employed to investigate the morphology and thickness of the obtained 2D MoP crystals. SEM image of MoP crystals (Fig. 1b) shows the hexagonal shape. From the EDX mapping (Fig. S2), P and Mo exhibit homogeneous distribution, while Ga is not detected. In addition, the corresponding EDX spectrum of MoP shows the counts of P and Mo are 100%, and the atomic ratio of P and Mo is about 1:1 (Table S1). These experimental results show the high purity of 2D MoP. EDX elemental mapping images collected from TEM of Mo and P for MoP crystal are also shown in Fig. 1d. The dark spots in EDS mapping might be caused by radiation damage of e-beam irradiation [21], which results in the damage of samples and appearance of some dark spots [22]. The thickness of MoP was detected by AFM, revealing a thickness of 10 nm (Fig. 1c). Controlling growth time can obtain 2D MoP single crystals with different thicknesses (Fig. S3). In addition, XPS was employed to survey the surface electronic state of each element in MoP crystals. The P 2p exhibits two distinct peaks observed at 129.2 and 130.1 eV, representing the P 2p_{3/2} and P 2p_{1/2} related to P bonded with Mo (Fig. S4a), respectively [23]. The Mo 3d spectrum exhibits the peaks at 231.1 and 227.9 eV, which can be designated as Mo in the MoP (Fig. S4b) [24]. XPS result of the fitted Ga 2p_{3/2} spectra is shown in Fig. S4c. The main peak centered at binding energy of 1118.9 eV results from the Ga₂O₃ at Ga surface. The peak located at 1118.2 eV is due to Ga₂O at

Ga surface. The peak located at 1116.8 eV corresponds to metal state Ga⁰ [25]. The results are coincident with the fact that Ga spontaneously forms an oxide layer under ambient conditions [26]. The peak of Ga–P (1117.3 eV) is not detected [27], also indicating the high purity of the prepared MoP. Moreover, Raman spectrum of MoP single crystals was performed with one notable peak at 409 cm^{−1} (Fig. S5). According to the calculation, the E mode is Raman active due to the absence of inversion symmetry in MoP crystal [28]. Raman spectrum of MoP single crystals is similar to that in the reported study [29].

Crystal structure and crystalline quality of the prepared MoP samples were evaluated by HRTEM and STEM. Low-magnification HAADF-STEM image of the obtained hexagonal MoP crystal is given in Fig. 2a. Selected area electron diffraction (SAED) collected on this sample in Fig. 2b demonstrates only one set of hexagonally arranged diffraction spots, confirming the hexagonal-symmetry atomic arrangement and single-crystalline characteristic of the MoP [30]. The typical atomic model of (001) plane from the hexagonal MoP with *D*_{3h} symmetry is displayed in Fig. 2c. Fig. 2d, e display the atomic-level HAADF-STEM image and BF-STEM image of the 2D MoP single crystal, respectively. In Fig. 2d, the brighter spots are the Mo atoms and the dimmer spots are the P atoms. The hexagonal lattice consisting of P and Mo atoms can be identified, which agrees well with the *D*_{3h} symmetry indicated in Fig. 2c. The nearly perfect atomic structure reveals a lattice spacing of 0.32 nm, in parallel with (001)

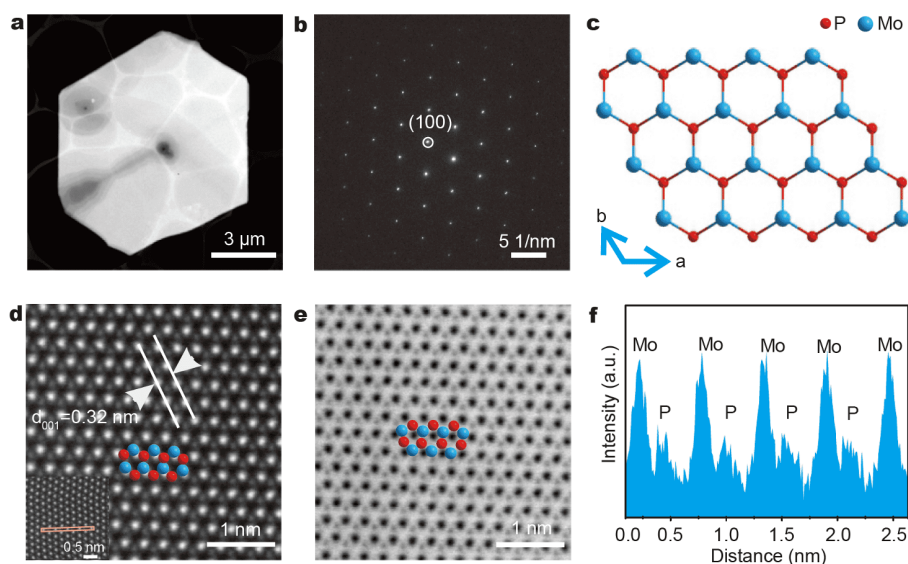


Figure 2 Characterization of atomic structure of 2D MoP single crystal. (a) Low-magnification HAADF-STEM image of a 2D MoP single crystal. (b) SAED pattern collected from the 2D MoP single crystal. (c) Scheme of the (001) face structure of MoP. (d, e) HAADF- and BF-STEM image of the 2D MoP single crystal, respectively. (f) Intensity profile of the selected area marked in (d) by a yellow rectangle.

planes in MoP [9]. Fig. 2f exhibits the intensity profile corresponding to the region marked in the inset of Fig. 2d by a yellow rectangle, which shows the intensity dispersion of Mo and P atoms. The uniform intensity differences between the bright and dim spots reveal that there are no vacancies or defects [31]. Furthermore, the HAADF intensity ratio of Mo and P is consistent with the value of $Z_{\text{Mo}}^{1.7}/Z_{\text{P}}^{1.7}$, which agrees well with the previous report [32]. Fig. 2e is the BF-STEM image of the 2D MoP single crystal and the brightness of the atoms is reversed, which can also clearly show the perfect hexagonal crystal structure of the 2D MoP. These characterizations indicate the as-prepared 2D MoP samples display high crystalline quality.

SHG refers to an NLO effect that two photons with the same frequency encounter with nonlinear materials producing one double frequency photon (Fig. 3a) [33]. The experimental setup of SHG measurement is shown in Fig. S6. We employed it to investigate the second-order NLO property and the potential applications in optoelectronics fields, and also to explore the structural information or crystal configuration of MoP single crystal. The nonlinear spectra of MoP were measured at different excitation wavelengths (Fig. 3b), showing that the SHG signal reaches a maximum value at the incident laser wavelengths of 860 nm. Excitation power dependence of SHG intensity was investigated at the incident laser with the power from 1 to 7.5 mW under the excitation wave-

length of 860 nm. In Fig. 3c, SHG signals can be observed at 430 nm displaying the second-order nonlinear process. Additionally, laser power dependence of SHG intensity is plotted in natural logarithm, displaying a fitting slope of 2.2 (Fig. 3d), close to the theoretically calculated value of 2, revealing the distinct quadratic relationship between the SHG intensity and excitation power [34]. Meanwhile, the homogeneous distribution of SHG mapping in Fig. 3e manifests the high quality and homogenous thickness of 2D MoP. We further studied the angle-dependent SHG intensity of MoP single crystal, because the polarization-resolved SHG depends on the structural symmetry. The structure of MoP belongs to the D_{3h} point group, the function of the angle-dependent SHG intensity can be theoretically predicted as $I = I_0 \cos^2(3\theta)$, where θ represents the azimuthal angle between the incident laser polarization and the mirror plane of a crystal, I and I_0 represent the detected SHG signal and the maximum SHG signal, respectively [35]. We observe a typical sixfold pattern with the azimuthal angle from 0° to 360° (Fig. 3f), further displaying the threefold symmetry of MoP crystal, in accordance with the HAADF-STEM image and the D_{3h} crystal structure.

To quantitatively investigate the SHG properties of MoP, we calculated the SHG susceptibility of MoP by using DFT calculation. Fig. 4a shows the dispersion of the NLO susceptibilities with different photon energies. Because the structure of MoP belongs to the D_{3h} point

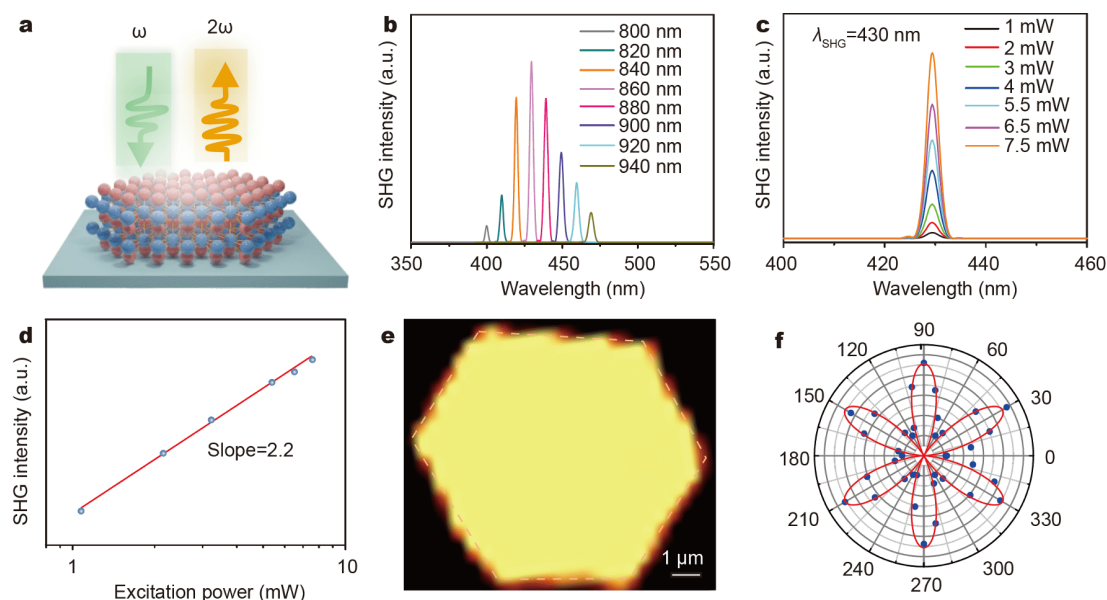


Figure 3 Investigation of SHG of the as-synthesized 2D MoP single crystal. (a) Schematic diagram of an SHG process. (b) Excitation wavelength-dependent SHG spectra. (c) Excitation power-dependent SHG spectra. (d) The corresponding linear fitting in logarithmic coordinates. (e) SHG mapping of a typical hexagonal MoP single crystal. (f) Polarization-angle-dependent SHG intensity of 2D MoP single crystal.

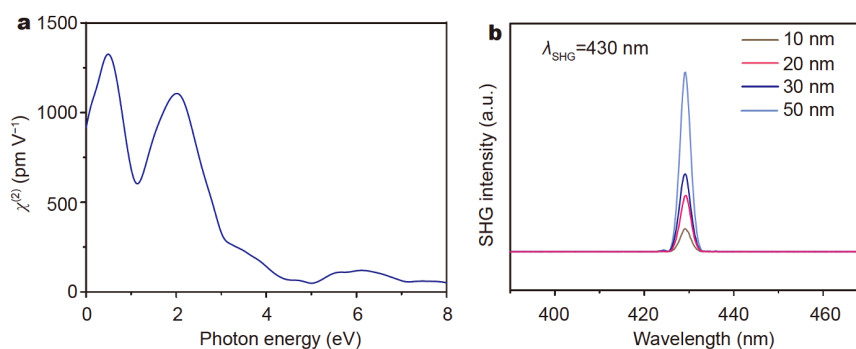


Figure 4 (a) Dispersion of the second-order nonlinear optical susceptibilities of MoP with different photon energies derived from DFT calculation. (b) SHG intensity of MoP single crystals with different thicknesses.

group, there is only one independent component [36]:

$$\chi^{(2)} = \chi_{xxi}^{(2)} = \chi_{xyx}^{(2)} = \chi_{yxx}^{(2)} = -\chi_{yyy}^{(2)}$$

The calculated SHG susceptibility of MoP at 860 nm is about 815 pm V^{-1} , which makes it a promising candidate as ultrathin frequency doubling crystal. We also compared the second-order NLO susceptibility of MoP with other reported typical 2D nonlinear optical crystals and commercially available nonlinear optical crystals (Table S2). The comparison results also indicate the high second-order NLO susceptibility of 2D MoP. In addition, the thickness-dependent SHG signals of 2D MoP at an excitation wavelength of 860 nm were investigated, as shown in Fig. 4b. SHG intensity of 2D MoP gets further

intensified as the sample thickness increases from 10 to 50 nm, which is consistent with the reported study [37]. Such a characteristic of MoP plays a crucial role in the NLO field since it enables a stable frequency conversion efficiency under different thicknesses [37].

CONCLUSIONS

In summary, we report the growth of ultrathin 2D MoP single crystals on liquid metals by CVD. Liquid Ga possesses an atomically smooth surface, which is one of the most suitable growth substrates to synthesize 2D materials. The obtained 2D MoP single crystals with a thickness of 10 nm display high quality. Next, we investigated the intrinsic strong SHG effect of MoP for the first time,

displaying that 2D MoP is the promising ultrathin frequency doubling crystal. Our study provides an approach to synthesize and explore many other 2D TMPs for future applications.

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Author contributions Fu L and Zeng M developed the concept and

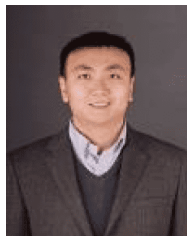
conceived the experiments. Cao F, Zheng S and Ding Y performed the main experiments. Liang J conducted the AFM characterization. Li Z conducted the DFT calculations. Wang Z and Wei B conducted the TEM characterization. Cao F wrote the manuscript. Fu L, Xu N and Zeng M revised the manuscript. All of the authors contributed to the data analysis and scientific discussion.

Conflict of interest The authors declare no conflict of interest.

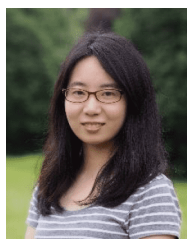
Supplementary information Supporting data are available in the online version of the paper.



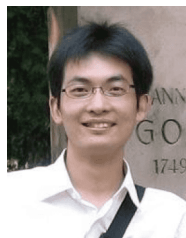
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Lei Fu received his PhD degree from the Institute of Chemistry, Chinese Academy of Sciences in 2006. After that, he worked as a Director's Postdoctoral Fellow at Los Alamos National Laboratory, Los Alamos, New Mexico (2006–2007). Thereafter, he became an Associate Professor at Peking University. In 2012, he joined Wuhan University as a full professor. His current interests of research relate to controlled growth and novel property exploration of 2D atomic layer thin crystals.

化学气相沉积法在液态金属上制备2D MoP单晶

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摘要 二维过渡金属磷化物(TMPs)有许多新奇的性质和应用. 作为二维TMPs的一员, 二维MoP有许多独特的物理化学性质. 然而由于缺乏制备二维MoP的方法, 目前还未成功制备二维MoP, 因此限制了对二维MoP众多性质的探索. 本文采用化学气相沉积法在液态金属镓(Ga)上制备了厚度为10 nm的二维MoP单晶. 液态Ga具有原子级平整的表面, 能作为制备二维材料的合适生长基底. 在生长过程中, Mo源扩散到Ga表面与磷源反应, 从而在Ga表面反应得到二维MoP单晶. 此外, 由于二维MoP具有本征的非中心对称结构, 文中首次研究了二维MoP的二次谐波信号的产生. 本文为其他二维TMPs的制备和性质探索提供了新思路.